

######

As structural materials, the green composite will be used under rain and moisture environments. For the application of structural material, there are many papers about the mechanical properties of water-absorbed green composite [16], [17]; but there are few papers about creep rupture strength and the life of water-absorbed green composite. This study examined the creep rupture property of water-absorbed green composite for long-term safety.

2 SPECIMENS

The polylactic acid (PLA) resin sheet (Ecodear 250-1B01CA, Toray Co., Ltd.) was used as a matrix. Plain woven jute fiber cloth knitting jute spun yarn was used as reinforcement. Fig. 1 shows plain woven jute fiber cloth knitting jute spun yarn. For fabrication of green composite, the molding method was the vacuum-compression molding method. The molding temperature was 190°C, the molding pressure was 1.7 MPa, and the holding time was 10 min. After that, the mold was cooled to room temperature.

The water absorption rate of green composite at 24 hours was 8%, when green composite had been immersed in distilled water at room temperature. Fiber volume fractions of the non-absorbed and 8% water-absorbed green composite were 42%. Fig. 2 shows a schematic drawing of the specimen for the quasi-static tensile and creep tests. Specimen sizes were 200 mm long, 10 mm wide and 1.0 mm thick.

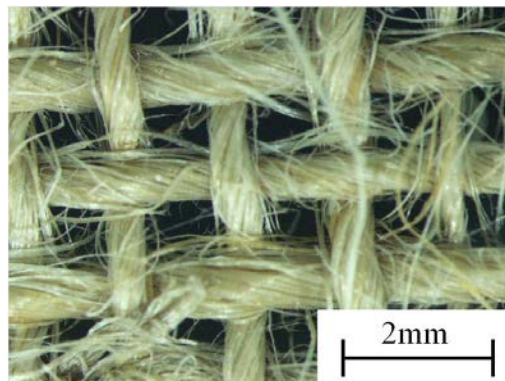


Figure 1: Plain woven jute fiber cloth.

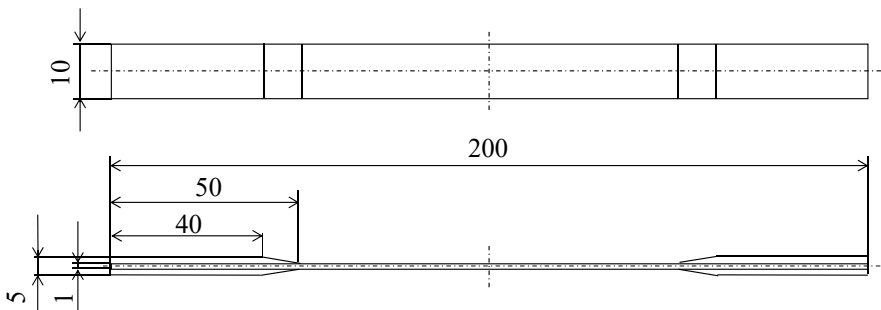


Figure 2: Schematic drawing of specimen sample for the quasi-static tensile and creep tests.

3 TESTING METHODS

3.1 Quasi-static tensile test

Quasi-static tensile tests of 8% water-absorbed green composites and non-water-absorbed controls were conducted based on the Japanese Industrial Standard (JIS) K 7164. The Tensilon RTG-1250A (A & D Co., Ltd.) was used as a tensile testing machine; at crosshead speeds of 0.1, 1.0, and 10 mm/min (strain rate: $3.4 \times 10^{-5} \text{ s}^{-1}$ - $3.4 \times 10^{-3} \text{ s}^{-1}$); and gauge length of 25 mm. Environmental temperature was room temperature. We tested five specimens.

3.2 Tensile creep test

Tensile creep test of 8% water-absorbed composite and non-water-absorbed controls were conducted, based on JIS K 7115. We used Creep tester 100LER (Toyo Seiki Seisaku-sho Co., Ltd.) as a testing machine. The maximum stress given was 50–90% of the tensile strength (crosshead speed: 1mm/min), at room temperature, and the maximum test time was 100 hours. We observed the fracture morphologies of 8% water-absorbed green composites and their non-water-absorbed controls by optical microscope (SZY7, Olympus Co., Ltd.) after the tensile creep test.

4 RESULTS AND DISCUSSION

4.1 Tensile property of green composite under strain rate

Figs 3 and 4 show Young's modulus and tensile strength of 8% water-absorbed green composites and non-water-absorbed controls under strain rates. Young's modulus and tensile strength of the 8% water-absorbed green composites and non-water-absorbed controls increased with an increase in strain rate. When the strain rate was $3.4 \times 10^{-3} \text{ s}^{-1}$, we found that Young's modulus and the tensile strength of 8% water-absorbed green composite decreased by 32% and 38%, respectively, as compared with the values obtained by non-water-absorbed green composite. Young's modulus and tensile strength of 8% water-absorbed green composite were lower than those of the non-water-absorbed green composite under all strain rates.

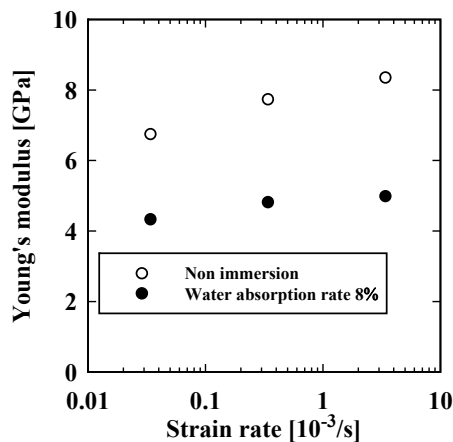


Figure 3: Young's modulus of water-absorbed green composites and non-absorbed controls under strain rates.

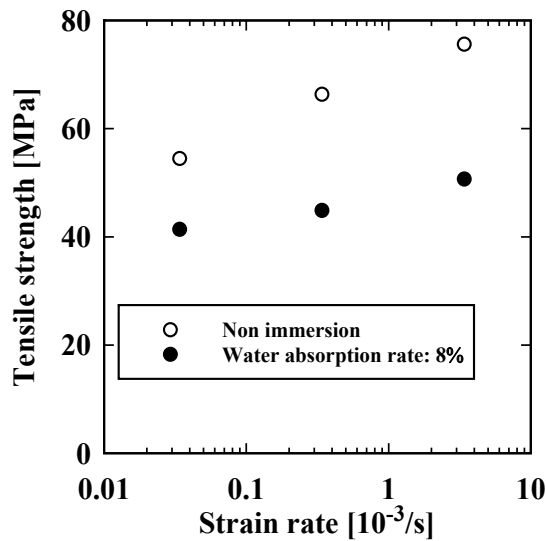


Figure 4: Tensile strengths of water-absorbed and non-water-absorbed green composites under strain rates.

Generally, the glass transition temperature of the PLA resin was decreased by water absorption [18]; thus, the strain rate dependence of the green composite was mainly affected due to a decrease of viscoelasticity by the water absorption of the matrix.

4.2 Creep rupture properties of water-absorbed green composites and their non-water-absorbed controls

Fig. 5 shows the creep rupture properties of 8% water-absorbed green composites and non-water-absorbed controls. Creep rupture strengths of control and 8% water-absorbed green composites decreased with an increase of loading time. The creep rupture strengths at 100 hours of control and 8% water-absorbed green composites were 33 MPa and 23 MPa, respectively. The creep rupture life of 8% water-absorbed green composite was lower than that of non-water-absorbed green composite. When the maximum stress was 33 MPa, the creep rupture life of the non-water-absorbed green composite decreased 99%, as compared with that of water-absorbed green composite. The creep rupture strength of the PLA resin decreased, with an increase of loading time [19]; thus, the creep rupture property of green composite probably decreased due to a decrease of the creep rupture property of the matrix, by water absorption.

Fig. 6 shows fracture morphologies of the non-water-absorbed (control) green composites after tensile creep tests. The fracture morphology of this non-water-absorbed green composite became smooth. There were no different fracture morphologies for non-water-absorbed green composites at maximum stresses of 60% and 80%. Also, no large yarn pull-outs occurred in the fracture morphologies of non-water-absorbed green composites at maximum stresses of 60% and 80%.

Fig. 7 shows fracture morphologies of 8% water-absorbed green composites after tensile creep tests. Fracture morphology of 8% water-absorbed green composite at a maximum

stress of 80% became smooth, but we found a large yarn pull-out in the fracture morphology of 8% water-absorbed green composite at a maximum stress of 60%. As mentioned above, the glass transition temperature of PLA resin was decreased by water absorption. The water probably penetrated into the interface between fiber and resin, when the green composite absorbed water. For these reasons, the yarn pull-out in the fracture morphology of green composite under a maximum stress of 60% probably occurred due to a decrease in interfacial adhesion between the fiber and resin; therefore, the creep rupture property of green composite was mainly affected due to a decrease in the viscoelasticity of the matrix by water absorption.

5 CONCLUSIONS

This study examined the creep rupture property of water-absorbed green composite for long-term safety. As a result, we concluded that: The tensile properties of non-absorbed and 8% water-absorbed green composites increased with an increase of the strain rate; the tensile properties of the 8% water-absorbed green composite were lower than those of non-water-absorbed green composite, for all strain rates; the strain rate dependence of green composite decreased by water absorption. Creep rupture strengths of 8% water-absorbed green composites and non-water-adsorbed controls decreased with an increase of loading time. The creep rupture life of 8% water-absorbed green composite was lower than that of non-water absorbed green composite. Fracture morphologies of 8% water-absorbed green composite (as well as the non-water absorbed) at maximum stress of 60% and 80% became smooth, except for the 8% water-absorbed green composite at a maximum stress of 60%; we found there was a large yarn pull out in the fracture morphology of 8% water-absorbed green composite at a maximum stress of 60%. Therefore, the creep rupture property of green composite was mainly affected, due to a decrease in viscoelasticity of the matrix by water absorption.

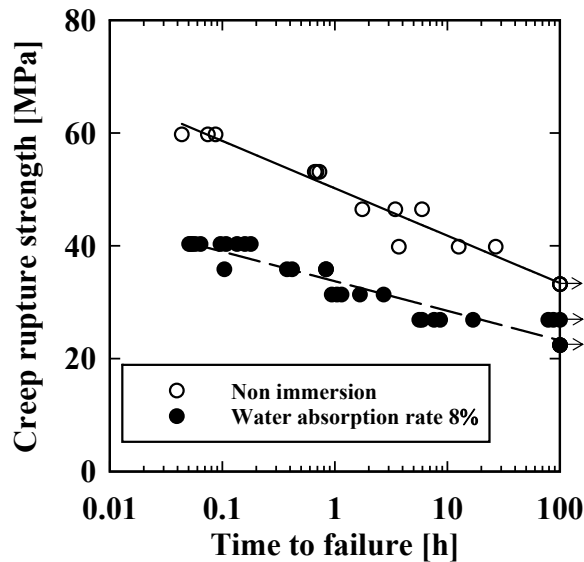


Figure 5: Creep rupture properties of 8% water-absorbed green composites and non-water absorbed controls.

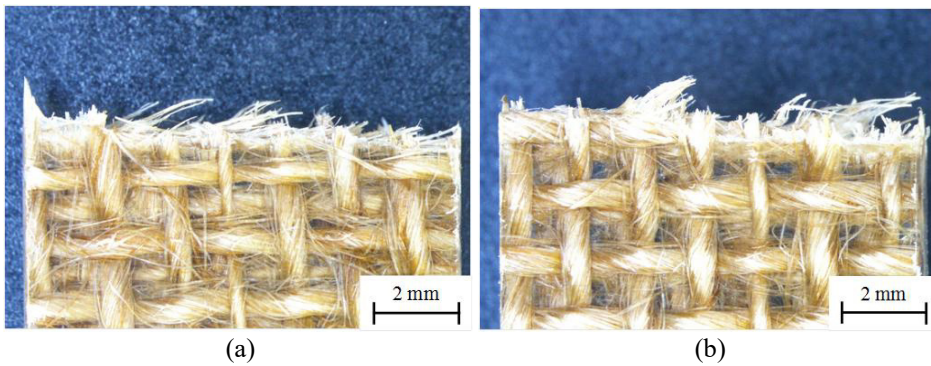


Figure 6: Fracture morphologies of the non-water-absorbed green composites after tensile creep tests. (a) Maximum stress 80 %; (b) Maximum stress 60 %.

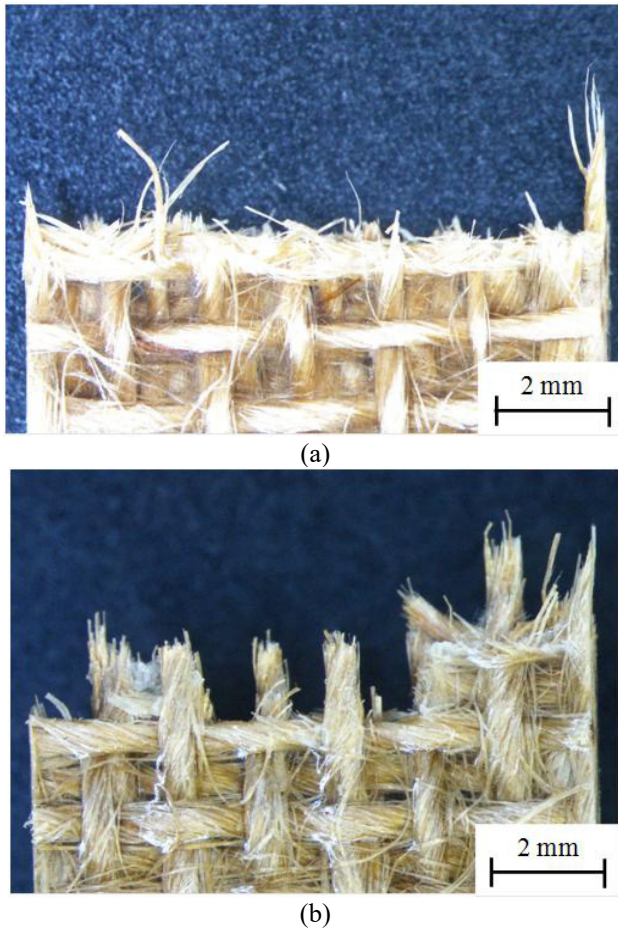


Figure 7: Fracture morphologies of 8% water-absorbed green composites after tensile creep tests. (a) Maximum stress 80%; (b) Maximum stress 60%.

REFERENCES

- [1] Koronis, G., Silva, A. & Fontul, M., Green composites: a review of adequate materials for automotive application. *Composites part B: engineering*, **44**, pp. 120–127, 2013.
- [2] Jayavani, S., Deka, H., Varghese, T.O. & Nayak, S.K., Recent development and future trends in coir fiber-reinforced green polymer composites: review and evaluation. *Polymer composites*, **37**, pp. 3296–3309, 2016.
- [3] Wei, L. & Mcdonald, A.G., A review on grafting of biofibers for biocomposites. *Materials*, **9**, pp. 303–326, 2016.
- [4] Toyota boshoku Co., Ltd, Online. <http://www.toyota-boshoku.com/jp/products/fiber-exterior/door-trim/index.html>.
- [5] Guapa Co., Online. <https://www.guapaco.com/pages/urban-one-black>.
- [6] Takagi, H. & Asano, A., Effects of Processing Conditions on Flexural Properties of Cellulose Nanofiber Reinforced “Green” Composites. *Composites part A: applied science and manufacturing*, **39**, pp. 685–689, 2008.
- [7] Kobayashi, S., Damage behavior of hemp fiber reinforced poly (l-lactic acid) composites under fatigue loading. *Journal of solid mechanics and materials engineering*, **7**, pp. 317–323, 2013.
- [8] Kobayashi, S., Takeda, K. & Nakamura, R., Processing and characterization of hemp fiber textile composites with micro-braiding technique. *Composites part A: applied science and manufacturing*, **59**, pp. 1–8, 2014.
- [9] Takagi, H., Nakagaito, A.N., Nishimura, K. & Matsui, T., Mechanical characterisation of nanocellulose composites after structural modification. *WIT Transactions on the Built Environment*, **166**, pp. 335–341, 2016.
- [10] Tanaka, K., Shiga, T. & Katayama, T., Fabrication of hydroxyapatite/PLA composite nanofiber by electrospinning. *WIT Transactions on the Built Environment*, **166**, pp. 371–379, 2016.
- [11] Katogi, H., Takemura, K. & Akiyama, M., Residual tensile property of plain woven jute fiber/poly(lactic acid) green composites during thermal cycling. *Materials*, **9**, pp. 573–583, 2016.
- [12] Ren, B., Mizue, T., Goda, K. & Noda, J., Effects of fluctuation of fibre orientation on tensile properties of flax sliver-reinforced green composites. *Composite structures*, **94**, pp. 3457–3464, 2012.
- [13] Katogi, H., Shimamura, Y., Tohgo, K., Fujii, T. & Takemura, K., Fatigue behavior of unidirectional jute spun yarn reinforced PLA. *Advanced composite materials*, **21**, pp. 1–10, 2012.
- [14] Alvarez, V.A., Kenny, J.M. & Vazquez, A., Creep behavior of biocomposites based on sisal fiber reinforced cellulose derivatives/starch blends. *Polymer composite*, **25**, pp. 280–288, 2004.
- [15] Takemura, K., Takada, Y. & Katogi, H., Effect of treatment using silane coupling agent on creep properties of jute fiber reinforced composites. *WIT Transactions on the Built Environment*, **124**, pp. 417–424, 2012.
- [16] Liu, W., Misra, M., Askeland, P., Drzal, L.T. & Mohanty, A.K., ‘Green’ composites from soy based plastic and pineapple leaf fiber: fabrication and properties evaluation. *Polymer*, **46**, pp. 2710–2721, 2005.
- [17] Nam, T.H., Ogihara, S., Nakatani, H., Kobayashi, S. & Song, J.I., Mechanical and thermal properties and water absorption of jute fiber reinforced poly (butylene succinate) biodegradable composite. *Advanced composite materials*, **21**, pp. 241–258, 2012.



- [18] Takeyama, Y. & Kawagoe, M., The dynamic viscoelasticity of molded PLA absorbing moisture and water. *Proceedings of 46th Hokuriku Shin'etsu branch meeting, the japan society of mechanical engineering*, pp. 11–12, 2009 (in Japanese).
- [19] Ooi, K. & Ohgi, J., Elongation behavior of poly lactic acid under creep loading. *Proceeding of the 59th JSMS annual meetings*, pp. 17–18, 2010 (in Japanese).

